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# Effect of SiC Particulate on Short Glass Fiber Reinforced Polyester Composite in Erosive Wear Environment

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#### Abstract

This study reports the effect on impact strength, physical properties and erosion behavior of silicon carbide (SiC) (0 wt%, 10 wt%, 20 wt%) filled short glass fibers (GF) on reinforced polymer composites. The effect of various operational variables, material parameters and their interactive influences on erosive wear behavior of these composites has been studied systematically. It is observed that silicon carbide (SiC) reinforced glass fiber reinforced polymer composites exhibit better erosion resistance as compared with unfilled polymer composite. However, in particulate filled composites the maximum erosion rate takes place at an impingement angle of 75  $^{\circ}$  and for unfilled composite the maximum erosion takes place at an impingement angle of 60  $^{\circ}$ . In this investigation, the physical and mechanical properties of all the unfilled and particulate filled composites are also evaluated for structural applications.

Keywords: Short E-glass fiber, silicon carbide, erosion wear, polymer composite

#### Introduction

Polymer composites are used in engineering applications such as aircraft components, automobiles, gears, pump impellers where the components undergo erosive wear. The popularity of polymers is due to their high specific strength and stiffness. However, polymer composites have poor erosion resistance as compared with metal matrix composites [1]. In fact, it is two to three orders of magnitude lower than metallic materials [2]. Hence, it is essential to evaluate their strength as well as their erosive behavior. To overcome such problems a second phase is introduced in the polymer to form a polymer matrix composite. To obtain the desired material characteristics for a particular application, it is important to know how the changes in performance characteristics of the composite vary with filler content under given loading conditions. In these places, fiber reinforced polymers have to function in severe erosive environments and this often leads to their failure. Hard particulate fillers consisting of ceramic or metal particles and fiber fillers made of glass are

being used these days to dramatically improve the wear resistance, even up to three orders of magnitude [3]. Fillers affect the strength of the according their packing composite to characteristics, size and interfacial bonding [4]. Srivastava and Shembekar [5] showed that the fracture toughness of epoxy resin could be improved by addition of fly ash particles as filler. Also the improved performance of polymers and their composites in tribological applications by the addition of filler materials has shown great promise and so has lately been a subject of considerable interest. Various kinds of polymers and polymer matrix composites reinforced with metal particles have a wide range of industrial applications, such as heaters, electrodes [6] composites with thermal durability at high temperature [7] etc. The inclusion of inorganic fillers into polymers for commercial applications is primarily aimed at cost reduction and stiffness improvement [8,9]. Along with fiber-reinforced composites, the composites made with particulate

fillers have been found to perform well in tribological conditions.

The possible studies of the erosion characteristics of polymer composites are required due to operational requirements in dusty environments. Therefore, it is necessary to have a full knowledge of all system variables on the wear rate in order to take appropriate steps in the design of machine or structural components, and also to choose such materials to reduce/control this wear mode. Generally, variables influencing the erosive wear of composite materials include mechanical properties of the composites, fiber loading, filler contents, eroding particle size, impingement angle and impact velocity. In viewing literature on erosive wear of composites, most efforts were focused on the study of the influence of the material properties rather than the operating parameters [10-13]. Several investigations on friction and wear properties of polyether ether ketone (PEEK) and its composites filled with fibers, organic, and inorganic fillers have been carried out [14,15]. Cirino et al. [14,16] reported the sliding as well as the abrasive wear behavior of continuous carbon and aramid fiber-reinforced PEEK.

Silicon carbide (SiC) is one such ceramic material that has the potential to be used as a filler in various polymer matrices. It was originally produced by a high temperature electrochemical reaction of sand and carbon. Today the material has been developed into a high quality technical grade ceramic with very good mechanical properties. It is used in abrasives, refractories, ceramics, and numerous high performance applications. The material can also be made an electrical conductor and has applications in resistance heating, flame igniters, and electronic components. Silicon carbide is composed of tetrahedral carbon and silicon atoms with strong bonds in the crystal lattice. This produces a very hard and strong material. It is not attacked by any acids, alkalis, or molten salts up to 800 °C. The high thermal conductivity coupled with low thermal expansion and high strength gives this material exceptional thermal shock resistant qualities. Silicon carbide has low density of about  $3.1 \text{ g/cm}^3$ , low thermal expansion, high elastic modulus, high strength, high thermal conductivity. high hardness, excellent thermal shock resistance, and superior chemical inertness [17,18].

In this study, physical properties (hardness and density etc.), impact strength and the erosive wear behaviors of SiC/ GFR filled polyester composites were examined. This study evaluates the influence of impingement angle and impact velocity on erosion wear behaviour of composites. The results indicated the effect of short glass fiber reinforcement and inclusion of SiC filler on the impact strength, physical properties and erosive wear behavior of the composites.

### Materials and methods

## **Preparation of composites**

Short glass fibres (Elastic modulus of 72.5 GPa, density of 2.59 gm/cc) (Twaron, Teijin) of 6 mm length were used to prepare the composites. The unsaturated isophthalic polyester resin (Elastic modulus 3.25 GPa, density 1.35 gm/cc) is manufactured by Ciba Geigy and locally supplied by Northern Polymers Ltd. New Delhi, India. The composites are made by conventional hand lay-up technique. Two percent Cobalt naphthenate (as an accelerator) is mixed thoroughly in isophthalic polyester resin and then 2 % methyl-ethyl-ketoneperoxide (MEKP) as a hardener is mixed in the resin prior to reinforcement. The filler material SiC was provided by NICE Ltd, India. Composites of three different compositions (0, 10 and 20 wt% SiC filling) are made and the fiber-loading (weight fraction of glass fiber in the composite) is kept at 50 wt% for all the samples. The castings are put under load for about 24 h for proper curing at room temperature. Specimens of suitable dimension are cut using a diamond cutter for different properties characterization and erosion testing.

### Physical characterization of composites

#### Density

The theoretical density of composite materials in terms of weight fraction can easily be obtained from the following equations given by Agarwal and Broutman [19].

$$\rho_{\rm ct} = \frac{1}{(W_f / \rho_f) + (W_m / \rho_m)}$$
(1)

where, W and  $\rho$  represent the weight fraction and density respectively. The suffix f, m and ct stand for the fiber, matrix and the composite materials respectively.

$$\rho_{\rm ct} = \frac{1}{(W_f / \rho_f) + (W_m / \rho_m) + (W_p / \rho_p)}$$
(2)

where, the suffix p indicates the particulate filler materials.

The actual density ( $\rho_{ce}$ ) of the composite, however, can be determined experimentally by simple water immersion. Therefore the volume fraction of voids (V<sub>f</sub>) in the composite is calculated using the following equation:

$$V_{f} = \frac{\rho_{ct} - \rho_{ce}}{\rho_{ct}}$$
(3)

#### **Micro-hardness**

be written as

Micro-hardness measurement is done using a micro-hardness tester equipped with a square based pyramidal (angle 136 ° between opposite faces) diamond indenter by applying a load of 10 N as per ASTM E-18.

#### Impact test

Low velocity instrumented impact tests were carried out on composite specimens. The tests were done as per ASTM D 256 using an impact tester. The pendulum impact testing machine ascertains the notch impact strength of the material by shattering the V notched (45 °) specimen with a pendulum hammer, measuring the spent energy, and relating it to the cross section of the specimen. The standard specimen for ASTM D 256 is 60 ×10  $\times$  4 mm<sup>3</sup> and the depth under the notch is 2 mm. The machine is adjusted such that the blade on the free-hanging pendulum just barely contracts the specimen (zero position). The specimens were clamped in a square support and were struck at their central point by a hemispherical bolt of diameter 5 mm. The respective values of impact energy of different specimens were recorded directly from the dial indicator.

#### Taguchi experiment design

For the elaboration of the experimental plan the method of Taguchi for five factors at three levels was used, being understood by levels taken by the factors. In **Table 1** is indicated the factors to be studied and the assignment of the corresponding levels. In the present study, we choose the experimental layout of the  $L_{27}$  (3<sup>13</sup>) orthogonal array consisting of five control factors, each one at three levels, as shown in **Table 2**. The factors and the interactions were assigned to the columns. In practice, these factors can be assigned arbitrarily to any of the arrays columns, provided that all combinations are included.

After assigning appropriate level settings, the S/N analysis (S/N: signal-to-noise ratio) was needed to evaluate experiment results. In S/N analysis, the greater the S/N, the better the experimental results:

$$\eta = -10 \log (M.S.D.)$$
 (4)

where M.S.D. is the mean-square deviation for the output characteristic (erosive wear rate).

As mentioned earlier, there are three categories of quality characteristics, i.e. lower-thebetter, higher-the-better, and nominal-the-better. To obtain optimal performance, lower-the-better characteristic for erosion rate must be taken. The mean-square deviation (M.S.D.) for the lower-thebetter characteristic can be expressed as [20]:

$$M.S.D = \frac{1}{m} \sum_{i=1}^{m} T_i^2$$
(5)

where *m* is the number of tests and  $T_i$  is the value of experimental result of the *i* th test.

The plan of the experiments was as follows: the first column was assigned to impact velocity (A), the second column to SiC filler content (B), the fifth column to impingement angle (C), the ninth column to stand-off distance (D) and the tenth column to erodent size (E), the third and fourth column were assigned to  $(A \times B)_1$  and  $(A \times B)_2$  respectively to estimate interaction between impact velocity (A) and SiC percentage (B), the sixth and seventh column were assigned to  $(B \times C)_1$  and  $(B \times C)_2$  respectively to estimate the interaction between the filler content (B) and impingement angle (C), the eight and eleventh column are assigned to  $(A \times C)_1$  and  $(A \times C)_2$ respectively to estimate interaction between the impact velocity (A) and impingement angle (C) and the remaining columns were used to estimate experimental errors. The output to be studied was the erosion rate (Er) and the tests were repeated twice corresponding to 54 tests, to allow the analysis of variance of the results. Furthermore, a statistical analysis of variance (ANOVA) was performed to identify the process parameters that were statistically significant. With the S/N and ANOVA analyses, the optimal combination of the process parameters can be predicted and their level of significance.

**Table 1** Levels for various control factors.

Control factor		Level					
Control factor	Ι	II	III	Units			
A:Velocity of impact	43	54	65	m/sec			
B:Filler content	0	10	20	%			
C:Impingement angle	30	60	90	degree			
D:Stand-off distance	65	75	85	mm			
E:Erodent size	250	350	450	μm			

### **Table 2** Orthogonal array for an $L_{27}$ (3<sup>13</sup>) Taguchi Design.

L27	1	2	3	4	5	6	7	8	9	10	11	12	13
$(3^{13})$	Α	В	$(\mathbf{A} \times \mathbf{B})_1$	(A×B) <sub>2</sub>	С	( <b>B</b> × <b>C</b> ) <sub>1</sub>	(B×C) <sub>2</sub>	(A×C) <sub>1</sub>	D	Е	(A×C) <sub>2</sub>		
1	1	1	1	1	1	1	1	1	1	1	1	1	1
2	1	1	1	1	2	2	2	2	2	2	2	2	2
3	1	1	1	1	3	3	3	3	3	3	3	3	3
4	1	2	2	2	1	1	1	2	2	2	3	3	3
5	1	2	2	2	2	2	2	3	3	3	1	1	1
6	1	2	2	2	3	3	3	1	1	1	2	2	2
7	1	3	3	3	1	1	1	3	3	3	2	2	2
8	1	3	3	3	2	2	2	1	1	1	3	3	3
9	1	3	3	3	3	3	3	2	2	2	1	1	1
10	2	1	2	3	1	2	3	1	2	3	1	2	3
11	2	1	2	3	2	3	1	2	3	1	2	3	1
12	2	1	2	3	3	1	2	3	1	2	3	1	2
13	2	2	3	1	1	2	3	2	3	1	3	1	2
14	2	2	3	1	2	3	1	3	1	2	1	2	3
15	2	2	3	1	3	1	2	3	3	1	2	3	1
16	2	3	1	2	1	2	3	3	1	2	2	3	1
17	2	3	1	2	2	3	1	1	2	3	3	1	2
18	2	3	1	2	3	1	2	2	3	1	1	2	3
19	3	1	3	2	1	3	2	1	3	2	1	3	2
20	3	1	3	2	2	1	3	2	1	3	2	1	3
21	3	1	3	2	3	2	1	3	2	1	3	2	1
22	3	2	1	3	1	3	2	2	1	3	3	2	1
23	3	2	1	3	2	1	3	3	2	1	1	3	2
24	3	2	1	3	3	2	1	1	3	2	2	1	3
25	3	3	2	1	1	3	2	3	2	1	2	1	3
26	3	3	2	1	2	1	3	1	3	2	3	2	1
27	3	3	2	1	3	2	1	2	1	3	1	3	2

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#### **Experimental set-up**

**Figure 1** shows the schematic diagram of the erosion test rig conforming to ASTM G 76. The set up was capable of creating reproducible erosive situations for assessing erosion wear resistance of the prepared composite samples. The erosion tester consists of an air compressor, an air particle mixing chamber and accelerating chamber. Dry compressed air was mixed with the particles which were fed at constant rate from a sand flow control knob through a convergent brass nozzle of 4 mm internal diameter. These particles impact the specimen which can be held at different angles

with respect to the direction of erodent flow using a swivel and an adjustable sample clip. The velocity of the eroding particles was determined using the double disc method [21].

In the present study, dry silica sand (spherical) of different particle sizes (250, 350, 450  $\mu$ m) were used as the erodent. The samples were cleaned in acetone, dried and weighed to an accuracy of  $\pm$  0.1 mg using a precision electronic balance. The process was repeated till the erosion rate attained a constant value called the steady state erosion rate.



Figure 1 Schematic diagram of the erosion test rig [22].

#### Scanning electron microscopy

The surfaces of the specimens were examined directly by a scanning electron microscope JEOL JSM-6480LV in the LV mode. The eroded samples were mounted on stubs with silver paste. To enhance the conductivity of the eroded samples, a thin film of platinum was vacuum-evaporated onto them before the photomicrographs were taken.

#### **Results and discussion**

# Effect of void fraction on filler content of the composite

The theoretical density values for unfilled and particulate filled glass polyester composites are calculated using the rules for mixtures and are presented in **Table 3**. These values are generally higher than the experimentally measured density values, also presented in **Table 3**. Reduction in the

density of composites compared to the expected values is attributed to the entrapment of air in the polyester matrix material in the process of mechanical mixing of short glass fibers in the resin. Thus the difference between densities calculated by these two techniques is a measure of voids and pores present in the composites. The entrapped air (referred to as voids) is undesired because it adversely affects the mechanical

properties of the composite. But when this glass fiber-reinforced polyester resin is filled with micro-sized SiC particles, the density of the resulting hybrid composites assumes higher values; that is, 1.61 g/cm<sup>3</sup> (void fraction 19%) and 1.72 g/cm<sup>3</sup> (void fraction 22 %) for composites with SiC contents of 10 and 20 wt%, respectively. Consequently, the void fraction in the composites increases with the filler content.

Table 3 Theoretical and experimental densities of the composites.

Sl. No.	Filler content (wt%)	Theoretical density (gm/cm <sup>3</sup> )	Experimental density (gm/cm <sup>3</sup> )	Void fraction (%)
1	0	1.30	1.19	8.4
2	10	1.61	1.29	19
3	20	1.72	1.34	22

#### Effect of hardness on filler content of the composite

It is seen that the hardness values of SiC-GFpolyester composites are found to be higher than the unfilled glass fiber-polyester composite as shown in Figure 2. It is observed from this study that hardness is improved with increasing in SiC contents in the glass polyester composite. The composite with 10 wt% SiC content is found to exhibit an increase in hardness as compared to the unfilled composite. For unfilled glass polyester composite the hardness value is 24 Hv but with the addition of 10 wt% and 20 wt% SiC filler contents in the composite this hardness value increases to 30 Hv - 34 Hv. It is further observed that there is large increase in the hardness value from unfilled glass polyester composite to 10 wt% filled glass polyester composite and only a minor increase in hardness from 10 wt% to 20 wt% filled glass polyester composite. In the case of the hardness test, a compression or pressing stress is in action. So the matrix phase and the solid filler phase would be pressed together and touch each other more significantly. Thus the interface can transfer pressure more effectively although the interfacial bond may be poor. This results in enhanced hardness.



Figure 2 Variation of Micro hardness with SiC filler percentage.

# Effect of impact energy on filler content of the composites

The impact property of a material is the capacity of the material to absorb and dissipate energy under impact or shock loading. Rothon [23] systematically studied mineral filled polymer composites mechanical properties and explained that the higher aspect ratio filler particulates needed to achieve stiffness in the composites and which inevitably cause increased stresses near the particulate edges in the polymer matrix ultimately causes failure under impact loads. Generally, the impact failure of a composite occurs by factors like matrix fracture, fiber/matrix debonding and fiber pullout. The impact energy values of different

composites recorded during the impact tests are given in **Figure 3**. It shows that the resistance to impact loading of the present composites improved with the addition of SiC fillers. For the composite without filler content the impact energy is 2.284 J and with the addition of SiC filler, from 10 wt% to 20 wt%, the impact energy increases from 3.0158 J to 4.0457 J respectively. Therefore, impact strength marginally increases in this study when the wt% of filler is increased from 10 wt% to 20 wt%. The suitability of a composite for such applications should therefore be determined not only by the usual design parameters, but by its impact or energy absorbing properties.



Figure 3 Variation of impact strength with SiC filler percentage.

#### **Steady state erosion**

Erosion wear behavior can be classified as ductile and brittle erosion although this grouping is not definitive because the erosion characteristics depend on the experimental conditions as much as on the composition of the target material. Thermoplastic matrix composites usually show ductile erosion while the thermosetting ones erode in a brittle fashion. It is well known that impingement angle is one of the most important parameters in the erosion process and for the ductile materials peak erosion normally occurs at  $15 - 20^{\circ}$  while for brittle materials the erosion damage is at a normal impact angle [10,11].

# Effect of impingement angle on filler content of the composites

The variation in erosion wear rate of the composites with impingement angle was studied, keeping all other parameters at fixed levels as shown in **Figure 4**. It is seen that the addition of SiC particles into short glass-polyester composite reduces the wear rate quite significantly. The unfilled glass polyester composite shows the peak erosion taking place at an impingement angle of 60 ° while for both the particulate filled composites (with 10 wt% and 20 wt% SiC content) peak erosion takes place at an impingement angle of 75 °. The composite filled with 10 wt% of SiC

shows the minimum erosion rate as compared to 20 wt% SiC filled composites. This is due to the proper mixing of the fiber and filler materials which takes place with the matrix material. Whereas, for unfilled glass polyester composites a maximum erosion rate is observed due to the absence of filler materials. Although in this study the polyester is ductile, the location of peak erosion has shifted to 60 ° from the usual 15 - 30 °, as it is reinforced with short glass fibers. This shift

in the erosion behavior is an indication of the loss of ductility and is obviously attributed to the brittle fibers. Further shifting of the impingement angle from 60 to 75 ° proves that the composites tend to become still more brittle with incorporation of SiC particles. The trend is similar for both the composites with SiC filler. Thus, it can be concluded that erosion performance of short glasspolyester composites improves with the addition of SiC particles.



Figure 4 Influence of impingement angle on erosion rates of SiC particulate filled composites.

# Effect of impact velocity on filler content of the composites

Similarly, the variation in erosion rate of unfilled and SiC filled composites with impact velocity is shown in **Figure 5**. Erosion trials were conducted at five different impact velocities. It is seen from **Figure 5** that for all the composite samples, the erosion rates gradually increase as impact velocity increases. The increase in erosion rate with impact velocity can be attributed to increased penetration of particles on impact as a result of dissipation of a greater amount of particle thermal energy to the target surface. This leads to more surface damage, enhanced sub-critical crack growth etc. and consequently to the reduction in erosion resistance.

## Taguchi analysis and response optimization

In **Table 4**, the last column represents S/N ratio of the erosion rate which is in fact the average of two reapplications. As an evaluation tool for determining robustness, the signal-to-noise (S/N) ratio is the most important component of a parameter design. The overall mean for the S/N ratio of the erosion rate is found to be -48.83 dB. The analysis is made using the popular software specifically used for design of experiment applications known as MINITAB 14. Before any attempt is made to use this simple model as a predictor, a measure of performance must be considered.

The significant effect of control factors level on erosion rate of the particulate filled glass polyester composites are shown in **Figure 6**. From **Figure 6** it is clear that if the level for a particular factor is nearer to the horizontal line, then the level of that factor has the least significant effect on the erosion rate of the composites [20,22]. On the other hand, a factor level for which the line has the highest inclination with the horizontal line will have the most significant effect on the output performance. Therefore, from this analysis it is

clear from **Figure 6** that factors  $A_2$  (Impact velocity at 54 m/sec), factor  $B_3$  (Filler content at 20 wt%), factor  $C_1$  (Impingement angle 30 °), factor  $D_1$  (stand-off distance 65 mm) and factor  $E_1$  (erodent size 250  $\mu$ m) respectively have been observed to be significant with erosion rate of the composites.



Figure 5 Influence of impact velocity on erosion rates of SiC particulate filled composites.





The interaction graphs are shown in the **Figures 7 - 9**. The interaction between impact velocity and filler content (A×B) is less significant on the output of the composites (**Figure 7**). From **Figure 6** the factor  $A_2$  and factor  $B_3$  are significant individually on output but the interaction effect of factor A and B on output may be less significant as shown in **Figure 7**. It is clear from **Figure 7** that the factor  $A_1$  and  $B_3$  line is at the higher side of  $A_2$ 

and  $B_3$ . Therefore, from this analysis it is clear that impact velocity (A) and filler content (B) have a significant effect on erosion rate independently but their combined effect is less significant. However, **Figures 8** and **9** show significant interaction effects on the erosion rate of the composites. The effect of the significance of control factors and interactions on output can be further studied by analysis of variance (ANOVA) in the next step.

Sl. No.	Impact Velocity (m/sec)	Filler content (wt%)	Impingement Angle (Degree)	Stand off Distance (mm)	Erodent size (µm)	Erosion rate (mg/kg)	S/N Ratio (dB)
1	43	0	30	65	250	272.26	-48.6997
2	43	0	60	75	350	318.66	-50.0666
3	43	0	90	85	450	324.8	-50.2323
4	43	10	30	75	350	407.46	-52.2017
5	43	10	60	85	450	344.53	-50.7445
6	43	10	90	65	250	271.73	-48.6828
7	43	20	30	85	450	250.66	-47.9817
8	43	20	60	65	250	145.06	-43.2310
9	43	20	90	75	350	226.66	-47.1075
10	54	0	30	75	450	308.03	-49.7719
11	54	0	60	85	250	315.46	-49.9789
12	54	0	90	65	350	321.6	-50.1463
13	54	10	30	85	250	219.46	-46.8271
14	54	10	60	65	350	252.8	-48.0555
15	54	10	90	75	450	280.26	-48.9512
16	54	20	30	65	350	133.6	-42.5161
17	54	20	60	75	450	278.66	-48.9015
18	54	20	90	85	250	246.4	-47.8328
19	65	0	30	85	350	436.8	-52.8057
20	65	0	60	65	450	432	-52.7097
21	65	0	90	75	250	376	-51.5038
22	65	10	30	65	450	304.8	-49.6803
23	65	10	60	75	250	395.46	-51.9421
24	65	10	90	85	350	286.4	-49.1395
25	65	20	30	75	250	128.8	-42.1983
26	65	20	60	85	350	240.53	-47.6234
27	65	20	90	65	450	284.8	-49.0908

**Table 4** Experimental design using an L<sub>27</sub> orthogonal array.



Figure 7 Interaction graph between  $A \times B$  for erosion rate.



Figure 8 Interaction graph between  $A \times C$  for erosion rate.

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Figure 9 Interaction graph between  $B \times C$  for erosion rate.

#### ANOVA and the effects of factors

In order to understand a concrete visualization of the impact of various factors like impact velocity (A), filler content (B), impingement angle (C), stand-off distance (D) and erodent size (E) and their interactions, it is desirable to develop an analysis of variance (ANOVA) table to find out the order of significant factors as well as interactions. Table 5 shows the results of the ANOVA with the erosion rate. This analysis was undertaken for a level of confidence of significance of 5 %. The last column of Table 5 indicates the main effects are highly significant.

From Table 5, one can observe that the filler contents (p = 0.009), erodent size (0.137), impact velocity (p = 0.229) and impingement angle (p =0.303) have great influence on the erosion rate. Similarly, the interaction of impact velocity/impingement angle (A×C) and filler content/impingement angle (B×C) combination are more significant than impact velocity×filler contents (p = 0.346) combination on erosion rate.

Source	DF	Seq SS	Adj SS	Adj MS	F	Р
А	2	10.504	10.504	5.252	2.18	0.229
В	2	93.826	93.826	46.913	19.49	0.009
С	2	7.857	7.857	3.929	1.63	0.303
D	2	7.561	7.561	3.780	1.57	0.314
Е	2	16.376	16.376	8.188	3.40	0.137
A×B	4	14.685	14.685	3.671	1.53	0.346
A×C	4	19.627	19.627	4.907	2.04	0.254
B×C	4	16.981	16.981	4.245	1.76	0.298
Error	4	9.628	9.628	2.407		
Total	26	197.046				

Table 5 ANOVA	table for	erosion rate.
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\*\*DF: degree of freedom, <sup>##</sup>Seq SS: sequential sum of squares, <sup>\$</sup>Adj. SS: extra sum of squares.
 <sup>\$\$</sup>Adj. MS: sequential mean squares, <sup>\*\*\*</sup>F: F-test, <sup>###</sup>P: percent contribution.

Figure 10 presents the surface morphology of the unfilled short glass-polyester composites under various test conditions. After the impact of silica sand on the surface of the material to be eroded, the material is chipped off from the surface of the sample and the glass fiber is clearly visible as shown in Figure 10a. The impact angle, erodent particles size, stand of distance and the impact velocity are set at 30 °, 350 µm, 75 mm and 43 m/s, respectively. The micrograph of the eroded surface with a higher magnification is presented in Figure 10b in which a crater is formed due to the removal of material. Due to the repeated impact of hard silica sand particles there is initiation of cracks on the fiber body and these cracks subsequently propagate on the matrix bodies both in a transverse as well as a longitudinal manner.

Crack initiation and propagation causes removal of the material. Similar observations are shown in the Figures 10c, 10d, where the material removal in the composites is dominated by micro ploughing and micro cutting. Figure 11 shows the micrographs of the eroded surfaces of short glasspolyester composites filled with different weight proportions of SiC particles. It is seen that the composite faces reveal several stages of erosion and material loss. Very small holes and initiation of short cracks are seen on the eroded surface of the composite with SiC (10 wt%), in Figure 11a, indicating the initiation of matrix material loss from the surface. The plastic deformation of the matrix material at lower impingement angle (30°), lower impact velocity (43 m/s) and at lower impact SiC contents (10 wt%), are also shown in Figures 11a and 11b.



Figure 10 SEM micrographs of the eroded surfaces of the unfilled glass-polyester composites.

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Figure 11 SEM micrographs of the eroded glass-polyester composites filled with SiC.

**Figure11c** shows the micrograph of the same composite surface eroded at higher impingement angle of 60  $^{\circ}$  and higher impact velocity 54 m/s. Therefore the matrix material which covers the fiber materials seems to be chipped off and crater formation takes place due to which the fibers are exposed to the environment. Also the roughing of

the surface takes place due to the repeated impact of erodent particles or hard silica sand particles. **Figure 11d** shows the fragmentation of fibers as a result of cracks and multiple fractures on the fiber body. The micrographs presented in **Figures 11e** and **11f** shows the SEM images for the glasspolyester filled with 20 wt% SiC contents. At an impact velocity (54 m/s) and impingement angle (60 and 90 °), the damage to the surface is minimal as seen in **Figures 11e** and **11f**. Subsequently the material removal becomes faster. In **Figure 11e** there is initiation of the cracks that takes place in the fiber after the removal of the matrix material. Formation of these cracks on the fiber body leads to the breakage of fibers in the composites and SiC filler fragmentation.

#### Conclusion

Based on the present study on mechanical and erosion behavior of SiC filled short glass fiber reinforced polyester composite the following observations and conclusions are made as:

(1) Successful fabrication of glass fiber and SiC reinforced polyester matrix composites is possible by simple hand lay-up techniques. Such composites have adequate potential for applications in highly erosive environments and their erosion wear performance shows significant improvement with the addition of SiC filler.

(2) The addition of SiC as a filler has a great effect on the impact strength and physical properties. The impact strength and hardness of the composites increases with increasing SiC filler contents in the glass polyester composites.

(3) In the case of unfilled glass-polyester composite the maximum erosion rate occurs at 60 ° impingement angle, whereas for particulate (10 wt% and 20 wt%) filled glass-polyester composite the erosion rate lies in between 70 to 80 ° impingement angle, due to the brittle nature of the glass fiber and reinforcement SiC particles. This indicates the semi ductile nature of the composites. This nature has been explained by analyzing the possible damage mechanism with the help of SEM micrographs. It is concluded that the inclusion of brittle fibers in the polyester matrix is responsible for this semi-ductility.

(4) Factors like erodent size, shape, stand off distance, SiC percentage and impingement angle, in order of priority, are found to be significant to minimize the erosion rate.

(5) Erosion characteristics of these composites can be successfully analyzed using a Taguchi experimental design scheme. The study reveals that for similar test conditions SiC filled glass polyester composites exhibit much better wear resistance compared to the unfilled glass polyester composite. This establishes that the addition of filler contents causes the erosion resistance of the composite to increase.

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